

# Manifestation of the Cooperative Jahn–Teller Effect in the Raman Spectra of $\text{Ba}_2\text{Cu}_x\text{Zn}_{1-x}\text{WO}_6$ Mixed Crystals

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Raman spectra of  $\text{Ba}_2\text{Cu}_x\text{Zn}_{1-x}\text{WO}_6$  mixed crystals ( $x = 0.0$  to  $0.4$  and  $x = 1$ ) are reported and shown to display considerable enhancement of Raman intensity for the  $\text{WO}_6$  vibration with  $E_g(O_h)$  symmetry when increasing the concentration of  $\text{Cu}^{2+}$ . Ab initio calculations of Raman frequencies and intensities for a  $\text{W}(\text{OH})_6$  model cluster indicate that the  $A_{1g}(D_{4h})$  component of this vibration may gain considerable intensity via mixing with the totally symmetric stretching  $A_{1g}(O_h)$  vibration when lowering the symmetry from  $O_h$  to  $D_{4h}$ , thus reflecting an indirect influence of Jahn–Teller distorted  $\text{CuO}_6$  centers on neighboring  $\text{WO}_6$  octahedra. Similarly striking is the change of the Raman spectrum in the energy region of the  $T_{2g}$  modes, when switching from the cubic phase with disordered local distortions in the  $\text{CuO}_6$ – $\text{WO}_6$  frame to the long-range cooperative Jahn–Teller order in  $\text{Ba}_2\text{CuWO}_6$ . It is concluded that Raman spectra can be used as a rather sensitive probe for detecting Jahn–Teller distortions in mixed crystals, thus providing a valuable addition to other spectroscopic techniques. © 1997 Academic Press

## I. INTRODUCTION

There has been convincing manifestations of the Jahn–Teller effect in stereochemistry and near IR, optical, and EPR spectroscopies of hexacoordinate  $\text{Cu}^{2+}$  (1,2). The  $^2E_g$  electronic ground state of octahedral  $\text{CuO}_6$  is electronically unstable and couples to vibrations of  $E_g$  symmetry,

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yielding absolute minima in the potential energy surface of the ground state with the typical pattern of tetragonal elongations along the three fourfold axes. In solids, elastic (lattice-mediated) interactions between distorted octahedra may lead to sterical correlations of the local distortions, which reduce the symmetry of the unit cell by lower symmetry cooperative strains and eventually lower one minimum with respect to the other two. At a given concentration of  $\text{Cu}^{2+}$  ions the strains which are random in a cubic phase may become oriented giving rise to a spontaneous lattice deformation (structural phase transition). An illustrative example for the cooperative Jahn–Teller effect are double perovskite (elpasolite)  $\text{Ba}_2\text{Cu}_x\text{Zn}_{1-x}\text{WO}_6$  mixed crystals ( $x = 0.0$ – $1.0$ ) (1,3). In  $\text{Ba}_2\text{CuWO}_6$  tetragonally elongated  $\text{CuO}_6$  octahedra [Cu–O bond distances:  $1.98 \text{ \AA}$  ( $4 \times$ ),  $2.42 \text{ \AA}$  ( $2 \times$ )] alternate with slightly tetragonally compressed  $\text{WO}_6$  octahedra [W–O bond distances  $1.96 \text{ \AA}$  ( $4 \times$ ) and  $1.90 \text{ \AA}$  ( $2 \times$ )], (Fig. 1), in contrast to cubic  $\text{Ba}_2\text{ZnWO}_6$  with regular octahedra of  $M^{\text{II}}\text{–O}$  and  $\text{W}^{\text{VI}}\text{–O}$  (bond distances of  $2.13$  and  $1.93 \text{ \AA}$ , respectively). When the  $\text{Cu}^{2+}$  concentration reaches a critical value,  $0.2$ , at  $300 \text{ K}$ , a structural phase transition takes place (Fig. 2a), lowering the symmetry from cubic ( $Fm\bar{3}m$ ) to tetragonal ( $I4/mmm$ ). It is accompanied by a discontinuous enhancement of the  $^2E_g$  ground state splitting which further increases monotonically with increasing  $\text{Cu}^{2+}$  content above the critical concentration—as evidenced from the  $d$ – $d$  transitions between  $^2B_{1g}$  ( $dx^2-y^2$  ground state) and  $^2A_{1g}$  ( $dz^2$ -excited state) observed in the near IR (see Fig. 2b). A phase diagram is shown in Fig. 2c. In this paper we present Raman spectra of mixed crystals  $\text{Ba}_2\text{Zn}_{1-x}\text{Cu}_x\text{WO}_6$  ( $x = 0.0$ – $1.0$ ) which

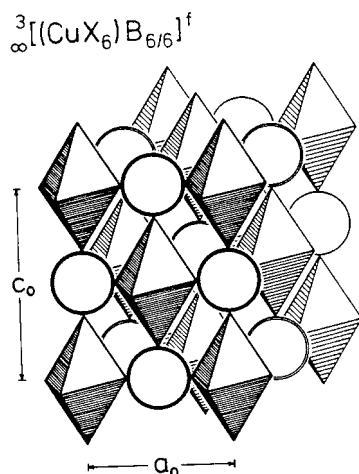


FIG. 1. The crystal structure of  $\text{Ba}_2\text{CuWO}_6$  with tetragonally elongated  $\text{CuO}_6$  octahedra (ferrodistorstive order). The  $\text{W}^{\text{IV}}$  ions are indicated by open circles (adopted from Ref. 1).

display interesting changes in dependence on  $x$ . Spectra, supplemented by ab-initio calculations of vibrational frequencies and Raman intensities are shown to provide a valuable tool for detecting the geometric changes around non-Jahn-Teller centers ( $\text{WO}_6$ ). Raman spectra of  $\text{Ba}_2\text{CuWO}_6$  are reported and shown to be particularly informative with respect to the Jahn-Teller active  $\text{CuO}_6$  clusters.

## II. EXPERIMENTAL SECTION

Sample preparations have been described previously (3). Raman spectra were recorded at 300 K, the technique being described elsewhere (4). They are interpreted on the basis of a factor group analysis within the  $Fm3m$  space group for the  $\text{Ba}_2\text{ZnWO}_6$  and the  $I4/mmm$  space group for the  $\text{Ba}_2\text{CuWO}_6$  solid. Raman frequencies, their relative intensities, and the changes with  $x$ , are rationalized by ab initio calculations, using effective core potentials of regular and tetragonally compressed octahedral  $\text{W}(\text{OH})_6$  model clusters and the ab initio program GAMESS (General Atomic and Molecular Electronic Structure System), developed by Schmidt *et al.* (5) and adopting the Stevens, Basch, and Krauss (SBK) basis sets (6). The calculations do not include adjustable (fitting) parameters. In order to study the influence of cooperative strains on the Raman vibrations we consider distortions of the  $\text{WO}_6^{6-}$  polyhedra and introduce one proton for each  $\text{O}^{2-}$  with a linear  $\text{W-O-H}$  geometry in order to counterbalance the excess negative charge. In a first step a geometry optimization ( $O_h$  symmetry) of the  $\text{W-O}$  and  $\text{O-H}$  bond distances has been done yielding bond distances of 1.868 and 0.951 Å, respectively. In a second step harmonic force constants ( $K_i$ ) and polarizability tensor components  $\alpha_{kl}$  ( $k, l = x, y, z$ ) have been calculated in dependence on displacements due to the normal modes  $Q_i$ , ( $i = 1$  to 5, Fig. 3)

$$\alpha_{kl} = \alpha_{kl}^0 + (d\alpha_{kl}/dQ_i)_0 Q_i. \quad [1]$$

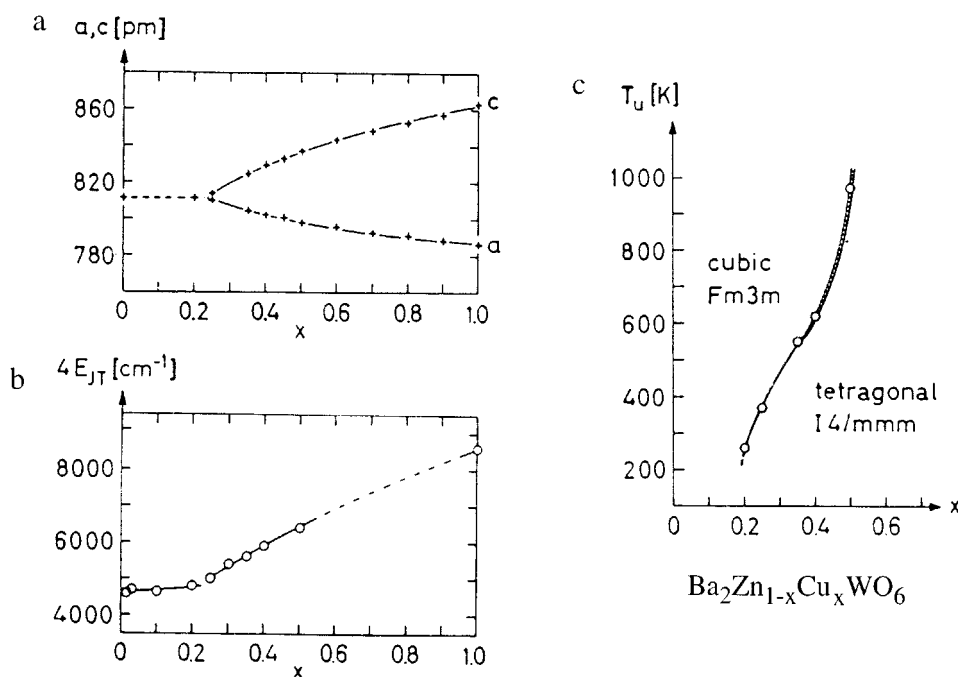


FIG. 2. The unit cell parameters (a), the  ${}^2E_g$  ground state splitting (b), and the transition temperatures from cubic to tetragonal (c) of mixed crystals  $\text{Ba}_2\text{Zn}_{1-x}\text{Cu}_x\text{WO}_6$  dependent on the  $\text{Cu}^{2+}$  concentration (adopted from Ref. 1).

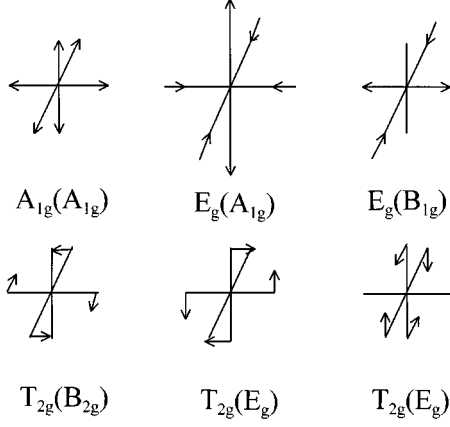


FIG. 3. The Raman active modes of octahedral  $\text{MO}_6$  polyhedra and their  $D_{4h}$  symmetry labels (in parentheses).

The intensities of the Raman transitions are approximated by

$$I(Q_i) \propto (d\alpha_{kl}/dQ_i)_0^2. \quad [2]$$

The same procedure has been followed when considering tetragonally compressed  $\text{W}(\text{OH})_6$  polyhedra. In tetragonal symmetry the  $\theta$  component of the octahedral  $E_g$  mode is totally symmetric and mixes with the  $A_{1g}(A_{1g})$  stretching vibration. The off-diagonal force field parameter  $K_{12}$

$$K_{12} = (d^2V/dQ_{A1}Q_\theta)_0 \quad [3]$$

defines the matrix which yields the mixing coefficients  $c_1$  and  $c_2$  of the normal modes  $A_{1g}(1)$  and  $A_{1g}(2)$

$$\begin{pmatrix} A_{1g}(A_{1g}) & A_{1g}(E_g) \\ K_{11} - K & K_{12} \\ K_{21} & K_{22} - K \end{pmatrix} \begin{pmatrix} c_1 \\ c_2 \end{pmatrix} = \begin{pmatrix} 0 \\ 0 \end{pmatrix}. \quad [4]$$

and the diagonalized force field parameters  $K$ :  $K_1$  and  $K_2$  in  $D_{4h}$ . The  $K_{ij}$  force constants are calculated using total energies taken from ab initio calculations. Finally, the polarizability tensor components of the  $D_{4h}$ -distorted octahedra are calculated using the same procedure as in  $O_h$ , allowing one to deduce the influence of geometrical distortions on the Raman frequencies and intensities.

### III. RESULTS

Raman spectra of  $\text{Ba}_2\text{Cu}_x\text{Zn}_{1-x}\text{WO}_6$  for  $x$  between 0.0 and 0.4 and of  $\text{Ba}_2\text{CuWO}_6$  are shown in Fig. 4. The spectra of the mixed crystals consist of three main lines, at 120, 430 and  $820 \text{ cm}^{-1}$  which positions do not change significantly

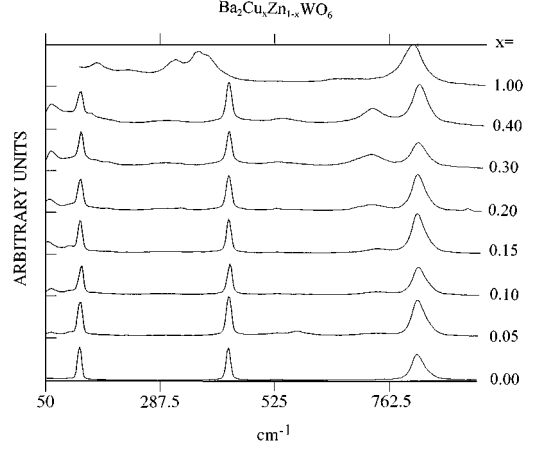


FIG. 4. The Raman spectra of  $\text{Ba}_2\text{Zn}_{1-x}\text{Cu}_x\text{WO}_6$  mixed crystals (at 300 K).

with  $x$  ( $0.0 < x < 0.4$ ). Two additional weak transitions are indicated in the high resolution spectrum of the  $\text{Ba}_2\text{ZnWO}_6$  host solid at  $530$  and  $750 \text{ cm}^{-1}$  (Fig. 5) and found to persist in the spectra of the  $\text{Cu}^{2+}$  doped samples. As the  $\text{Cu}^{2+}$  concentration increases the intensity of the  $750 \text{ cm}^{-1}$  transition grows and a further transition appears at  $55 \text{ cm}^{-1}$ . Passing to the  $\text{Ba}_2\text{CuWO}_6$  we note a drastic change in the Raman spectrum, both with respect to the intensities and the positions of the Raman bands (Fig. 6). The  $820 \text{ cm}^{-1}$  line slightly shifts to lower energies and seems to integrate the former  $750 \text{ cm}^{-1}$  transition, giving rise to a rather broad band, while an additional weaker component is detected at  $664 \text{ cm}^{-1}$ . The Raman lines for the mixed crystals at  $120$  and  $430 \text{ cm}^{-1}$  are significantly shifted and split by about  $65 \text{ cm}^{-1}$ , yielding two pairs of transitions at  $152, 220 \text{ cm}^{-1}$  and  $308, 370 \text{ cm}^{-1}$ . In order to interpret the experimental results a factor group analysis of the lattice vibrations has been performed for the  $\text{Ba}_2\text{ZnWO}_6$  ( $Fm3m$

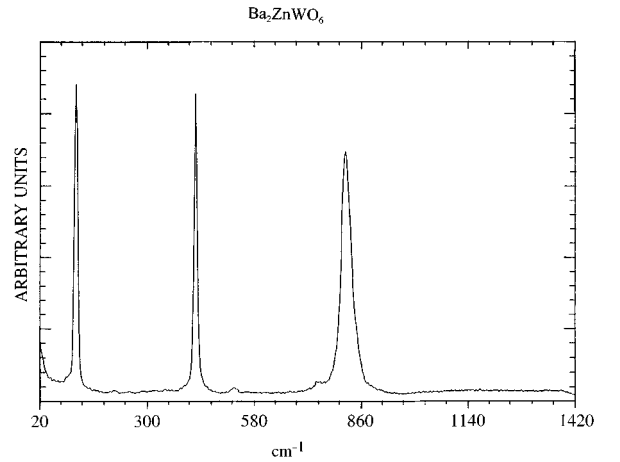


FIG. 5. The Raman spectrum of  $\text{Ba}_2\text{ZnWO}_6$  (at 300 K).

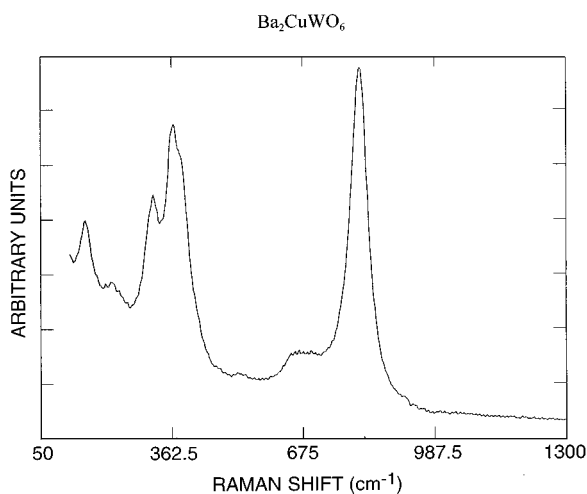


FIG. 6. The Raman spectrum of  $\text{Ba}_2\text{CuWO}_6$  (at 300 K).

space group). The atomic displacements for the Zn(W), Ba, and O ions occupying sites of  $O_h$ ,  $T_d$ , and  $C_{4v}$  point symmetries, respectively, are integrated into ( $O_h$ ) factor group species, to give four Raman active modes with  $A_{1g}$ ,  $E_g$ , and  $T_{2g}$  ( $2 \times$ ) symmetry (Table 1). Theoretical calculations on  $\text{W}(\text{OH})_6$  model clusters (see below) allow to assign these

TABLE 1  
Factor Group Analysis for  $\text{Ba}_2\text{ZnWO}_6$

Ions	Site symmetry (species)	$O_h$ factor group symmetry (species)
Zn(W)	$O_h$	$O_h$
	$T_{1u}$	$T_{1u}$
Ba	$T_d$	$O_h$
	$T_2$	$T_{2g}$
		$T_{1u}$
O	$C_{4v}$	$O_h$
	$A_1$	$A_{1g}$
		$E_g$
		$T_{1u}$
	$E$	$T_{1g}$
$T_{2g}$		
	$T_{2u}$	

Note. Space group symmetry:  $Fm\bar{3}m$ . Number of molecules in the unit cell  $Z = 4$ . Number of molecules in the Bravais space = 1.

$$\begin{aligned} \Gamma_{\text{crystal vibrations}} &= \Gamma_{\text{Ba}} + \Gamma_{\text{Zn}} + \Gamma_{\text{W}} + \Gamma_{\text{O}} - \Gamma_{\text{acoust}} \\ &= 4T_{1u}(\text{IR}) + 2T_{2g}(\text{R}) + A_{1g}(\text{R}) + E_g(\text{R}) + T_{1g} + T_{2u}. \end{aligned}$$

modes to the Raman lines at 820, 750, 430, and 120  $\text{cm}^{-1}$ , respectively. The two lines at higher energies (750 and 820  $\text{cm}^{-1}$ ) are predominantly attributed to vibrations of the  $\text{WO}_6$  cluster. Symmetry lowering from the  $O_h$  ( $\text{Ba}_2\text{ZnWO}_6$ , space group  $Fm\bar{3}m$ ) to the  $D_{4h}$  factor group ( $\text{Ba}_2\text{CuWO}_6$ , space group  $I4/m\bar{3}m$ ) leads to splittings ( $E_g$  into  $B_{1g}$ ,  $A_{1g}$ , and  $T_{2g}$  into  $B_{2g}$ ,  $E_g$ ) and energy shifts with the tentative assignment of the Raman lines at 152, 220, 308, 370, 664  $\text{cm}^{-1}$  to  $E_g$ ,  $B_{2g}$ ,  $B_{2g}$ ,  $E_g$ ,  $B_{1g}$  fundamentals, respectively, and of the broad band at 806  $\text{cm}^{-1}$  to the  $A_{1g}$  modes (From  $E_g$  and  $A_{1g}$  in  $O_h$ ). The origin of the bands at 55 and 530  $\text{cm}^{-1}$  is not fully understood. These bands are possibly due to correlated vibrations of the  $\text{CuO}_6$  and  $\text{WO}_6$  moieties, involving predominantly the  $\text{CuO}_6$ - and the  $\text{WO}_6$ -polyhedra, respectively.

#### IV. DISCUSSION

The intensity increase of the 750  $\text{cm}^{-1}$  Raman transition with increasing  $x$  can be led back to the Jahn–Teller properties of the  $\text{Cu}^{2+}$  cations and hence their tendency to distort the next-nearest surrounding. As far as these geometrical changes take place within the cubic symmetry (see the phase line in Fig. 2c,  $0.0 < x < 0.2$ , 300 K) a distorted surrounding according to elongated octahedra around  $\text{Cu}^{2+}$  will necessarily impose local distortions on the  $\text{WO}_6$  polyhedra in the opposite direction. In order to analyze the effect of such distortions on the Raman energies and intensities we performed calculations of the Raman frequencies and polarization tensors for an octahedral  $\text{W}(\text{OH})_6$  model cluster adopting  $\text{W–O}$  (1.868 Å) and  $\text{O–H}$  (0.951 Å) bond distances, optimized using a SBK basis, in comparison with analogous results for a tetragonally compressed  $\text{W}(\text{OH})_6$  octahedron [ $\text{W–O}$  bond lengths: axial 1.828 Å ( $2 \times$ ) and equatorial 1.888 Å ( $4 \times$ )]. The OH group has been approximated as a rigid moiety. A set of calculations in which the O–H group was considered as nonrigid did not change the results. The Raman active modes  $A_{1g}$ ,  $E_g$ , and  $T_{2g}$  of an isolated octahedral  $\text{MO}_6$  cluster are depicted in Fig. 3. Raman frequencies, polarizability tensor components  $\alpha_{kl}$  and relative intensities of the Raman lines, deduced from the relation in Eq. [2], are listed in Table 2. The results indicate that, for an undistorted  $\text{W}(\text{OH})_6$  octahedron,  $A_{1g}$  should be the most intense transition and  $E_g$  of the weakest intensity. Symmetry lowering to  $D_{4h}$  induces striking intensity changes only for the  $A_{1g}(2)$  component of the parent octahedral  $E_g$  vibration, gaining intensity via mixing with the  $A_{1g}(1)$  breathing vibration. The extent of intermixing depends on the magnitude of polyhedron distortion which becomes larger as the  $\text{Cu}^{2+}$  concentration increases. Therefore  $E_g$  vibrations mainly confined to  $\text{WO}_6$  clusters reflect in an indirect way structural changes due to the replacement of  $\text{Zn}^{2+}$  by  $\text{Cu}^{2+}$  in neighboring  $\text{ZnO}_6$  polyhedra. The strain imposed by Jahn–Teller distorted  $\text{CuO}_6$  centers on

**TABLE 2**  
**Calculated Raman Frequencies, Polarizability Tensor Components, and Intensities for Isolated Octahedral ( $O_h$ ) and Distorted Octahedral  $D_{4h}$  (Tetragonally Compressed)  $\text{W}(\text{OH})_6$  Model Clusters**

Raman active normal modes $Q$ ( $O_h$ symm.)	Raman energies ( $\text{cm}^{-1}$ )	Polariz. tensor components $\alpha_{ij}$ (a.u.)	Raman intensities (a.u./ $\text{\AA}^2$ )	Raman active normal modes $Q$ ( $D_{4h}$ symm.)	Raman energies ( $\text{cm}^{-1}$ )	Polariz. tensor components $\alpha_{ij}$ (a.u.)	Raman intensities (a.u./ $\text{\AA}$ )
$A_{1g}$	856	$\alpha_{xx,yy,zz}$ ( $59 - 25Q\alpha_{1g}$ )	625	$A_{1g}$ (1)	877	$\alpha_{xx,yy}$ ( $59 + 24Q\alpha_{1g}$ )	576
						$\alpha_{zz}$ ( $58 + 25Q\alpha_{1g}$ )	625
$E_g$	670	$\alpha_{xx}$ ( $59 + 4Q\epsilon_g$ ) $\alpha_{yy}$ ( $59 - 4Q\epsilon_g$ )	16	$A_{1g}$ (2)	695	$\alpha_{xx,yy}$ ( $59 - 11Q\alpha_{1g}$ )	121
			16				$\alpha_{zz}$ ( $58 - 0.1Q\alpha_{1g}$ )
				$B_{1g}$	631	$\alpha_{xx}$ ( $59 + 2.5Q\beta_{1g}$ )	6
						$\alpha_{yy}$ ( $59 - 3.5Q\beta_{1g}$ )	12
$T_{2g}$	794	$\alpha_{xy,xz,yz}$ ( $15Q\tau_{2g}$ )	225	$B_{2g}$	794	$\alpha_{xy}$ ( $15Q\beta_{2g}$ )	225
						$E_g$	794

*Note.* Adopted W–O and O–H bond lengths for the octahedral  $\text{W}(\text{OH})_6$  cluster are those obtained by a geometry optimization using effective core potentials (SBK basis);  $R(\text{W–O}) = 1.868 \text{ \AA}$  and  $R(\text{O–H}) = 0.951 \text{ \AA}$ . Data for  $D_{4h}$  refer to a tetragonally compressed octahedron with  $R(\text{W–O})_{ax} = R(\text{W–O}) - 0.04 \text{ \AA}$ ,  $R(\text{W–O})_{eq} = R(\text{W–O}) + 0.02 \text{ \AA}$  and with an unaltered O–H distance.

neighboring  $\text{WO}_6$  polyhedra in the cubic phase ( $x \leq 0.2$ ) becomes larger in the tetragonal phase ( $x = 0.3, 0.4$ ) leading to a remarkable intensity increase of the  $A_{1g}(2)$  mode in the latter case. The calculated Raman energies for the  $A_{1g}$  and  $E_g$  modes of the  $\text{W}(\text{OH})_6$  octahedron roughly match with those observed at 820 and 750  $\text{cm}^{-1}$ , indicating that these vibrations are nearly completely confined to the  $\text{WO}_6$  entities in the  $\text{Ba}_2\text{Cu}_x\text{Zn}_{1-x}\text{WO}_6$  mixed crystals. The  $T_{2g}$  mode at 430  $\text{cm}^{-1}$  seems to be strongly coupled with vibrational motions of the neighbored  $\text{Zn}(\text{Cu})\text{O}_6$  polyhedra, however—hence appearing at considerably lower energy than calculated (794  $\text{cm}^{-1}$ ; Table 2).

The Raman spectra of the mixed crystals ( $x \leq 0.4$ ) reflect a redistribution of Raman intensity from the  $A_{1g}(1)$  to the  $A_{1g}(2)$  mode of  $E_g$  due to the symmetry lowering from  $O_h$  to  $D_{4h}$ , without altering the band maxima positions significantly. In contrast, the spectrum of  $\text{Ba}_2\text{CuWO}_6$  displays a strongly changed Raman pattern involving both energy shifts and line splittings. These reflect the increase of the local distortion of the  $\text{WO}_6$  octahedra when moving from  $\text{Ba}_2\text{ZnWO}_6$  to  $\text{Ba}_2\text{CuWO}_6$ , caused by specific changes of the elastic properties of the bridging oxygen ions by the long-range cooperative ordering pattern (2). The enhanced splitting of the  $E_g$  vibration leads to a near-coincidence of the  $A_{1g}(2)$  split mode with  $A_{1g}(1)$ . The assignment of the

Raman pairs to the tetragonally splitted  $T_{2g}$  vibrational states at 152  $\text{cm}^{-1}$  ( $E_g$ ), 220  $\text{cm}^{-1}$  ( $B_{2g}$ ), and 308 ( $B_{2g}$ ), 370  $\text{cm}^{-1}$  ( $E_g$ ), which we attribute to elastically coupled symmetrized displacements from predominantly  $\text{CuO}_6$  octahedra (elongation) and from mainly  $\text{WO}_6$  octahedra (compression), respectively, reflect the different sign of the octahedral distortions around Cu and W. It should be reminded here that the tetragonal  $B_{2g}$  component of the parent octahedral vibrational state  $T_{2g}$  involves equatorial ligands only, while the  $E_g$  components include angular distortions with participation from both equatorial and axial M–O bonds (see Fig. 3). The intensity relations (see Fig. 6) support the assignment, because an  $E_g$  transition is expected to have (by statistical reasoning) about twice the intensity of  $B_{2g}$ . It is also in agreement with the proposed assignment, that the centre of gravity of the two  $T_{2g}$  states is nearly identical for the mixed crystals ( $x \leq 0.4$ ), namely 275  $\text{cm}^{-1}$ , and for  $\text{Ba}_2\text{CuWO}_6$  (262  $\text{cm}^{-1}$ ), though the energies of the  $T_{2g}$  split modes lie much closer in the latter case.

#### IV. CONCLUSION

1. The Raman spectra of  $\text{Ba}_2\text{Cu}_x\text{Zn}_{1-x}\text{WO}_6$  mixed crystals are found to reflect an interesting dependence of the

intensity of the Jahn–Teller active octahedral  $E_g$  mode on the  $\text{Cu}^{2+}$  concentration. The gain of intensity for the  $A_{1g}(E_g)$  split component—involving predominantly compressed  $\text{WO}_6$  octahedra ( $750\text{ cm}^{-1}$ )—can be used as a rather sensitive probe for detecting structural changes due to the vibronically active  $\text{Cu}^{2+}$  ion, which induces environmental distortions of neighbouring non-Jahn–Teller ions ( $\text{W}^{6+}$ ) as well.

2. A consistent assignment of the Raman lines observed at 150 and  $430\text{ cm}^{-1}$  ( $x \leq 0.4$ ) is possible on the basis of coupled bending motions of the  $\text{WO}_6$  and  $\text{Zn}(\text{Cu})\text{O}_6$  octahedra with  $T_{2g}$  symmetry. The transition from local distortions in the octahedral framework of the cubic elpasolite lattice due to the incorporation of  $\text{Cu}^{2+}$  to the long-range cooperative Jahn–Teller order in tetragonal  $\text{Ba}_2\text{CuWO}_6$  is accompanied by drastic changes in the spectral appearance (Figs. 4–6).

## ACKNOWLEDGMENTS

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